

# Universality of Superconducting Transition Temperatures

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## Abstract

It is hypothesized that superconducting transition temperatures are universal properties, which are amenable to a simple theoretical ansatz. The conjecture is supported by measured superconducting transition temperatures of many high quality samples. The experimental data contradict the mainstream view that an interaction between electrons and lattice vibrations is responsible for superconductivity.

**Keywords:** Superconductivity; transition temperature; universality; fundamental constants; number constants.

Based on various arguments, the author has calculated a heat energy to which all quantum particles of the vacuum couple. The temperature corresponding to this energy depends only on natural constants and is defined [1a] by  $k_B\tau \equiv 2^{-17/4} (hc)^{3/4}$ , where  $h$  is Planck's constant,  $c$  is the velocity of light in vacuum and  $k_B$  is Boltzmann's constant, which converts energy (J) into temperature (K). To get a unit consistent result, the kilogram (kg) must be replaced by  $\text{ms}^2$ . Based on the above definition, a value of  $\approx 1133$  K results for  $\tau$ , which can also be used to calculate the cosmic microwave background temperature [2].

Like the exotic fractional quantum Hall effect, superconductivity is a cooperative phenomenon of a system of electrons. Both phenomena are not understood at the deepest level. It is conjectured that in both cases the origins rely on strong correlations of electrons. The fractionalization of the Hall resistance as a function of a magnetic field is a well-confirmed experimental fact. The physical origin behind this universal effect could also play a role in other areas of physics, such as the transition of a material from the normal state to the superconducting (zero resistive) state. Science has frequently shown that analogies can unify and thus simplify the description of seemingly different phenomena.

The derivation of  $\tau$  is rooted on the nine-dimensional hypersphere. For this reason, 55 relatively prime fractions  $v \equiv p/q$  were formed with the numbers  $p, q \in \{1, 2, \dots, 9\}$ . The simple ansatz  $T_c = v \kappa \tau$  should apply to the critical temperature  $T_c$ , where the factor  $\kappa$  is a dimensionless measure of the pairing mechanism numerically specified by the numbers  $(2\pi)^{-1}$ ,  $\alpha/2$  or  $\alpha^2$ . The parameter  $\alpha$  is the Sommerfeld fine-structure constant defined by the number constant  $\alpha \equiv 2^{-6}\pi^{-2/3}$  [1b]. Putting everything together results in the set of critical temperatures  $T_c$  listed in ascending order in Table 1. The ansatz implies that superconductivity is driven by a universal mechanism that is not connected with electron-phonon interactions, but with the formation of electron pairs due to electromagnetic correlations.

Yang et al. 2000 [3] found that the correlation between  $T_c$  and the doping level evolved from a continuous parabola to discontinuous steps after oxidizing the single crystals. Table 1 may offer a rationale for explaining the observed phenomenon. Chu et al. 2015 [4] tabulated 80 transition temperatures  $T_c^{\max}$  that have been achieved for various compound classes of high temperature ( $T_c \geq 20$  K) superconductors. Assuming that the accuracy of the observed critical temperatures is  $\pm 1$  K, 58% of the values of Chu et al. corresponds to values in Table 1. Data for 43 optimally doped, high temperature superconducting compounds listed by Harshman et al. 2015 [5] might provide another clue of the pattern of occurrence. In this case, 70% corresponds to values in Table 1 and support the notion that superconductivity relies on universal ingredients. It is worth mentioning that optimally doped materials are not necessarily homogeneous materials with sharp transitions.

The determination of  $T_c$  is fraught with many hurdles because poly crystals and single crystals inevitably contain some disordered domains, that is, contain inclusions of secondary phases, stacking faults or impurities. All this determines the  $T_c$  and the associated width of the superconducting transition. Ideal behavior is rare except for exceptionally pure single crystals. The  $T_c$  is not only sensitive to inhomogeneities of the sample that may span nanometer to mesoscopic scales, but also, to a precise determination and regulation of the temperature in the measurements. It is also important how the transition itself is determined from experimental data, and how well the thermal equilibrium between the thermometer and the sample is settled. An analysis of the existing literature is challenging and each compilation must be examined critically.

In order to support the hypothesis, the author collected data on critical temperatures of samples with different morphologies. The data are summarized without comments for “pure” elements in Table 2, for binaries in Table 3, for complex compounds in Table 4 and for high temperature superconductors ( $T_c \geq 20$  K) in Table 5. It is up to the reader to judge to what extent these experiments support the conjecture  $T_c = v \kappa \tau$  and encompass the critical temperatures with a unified description. Hopefully, additional data obtained on high purity, strain-free “perfect” specimens with sharp superconducting transitions ( $\Delta T_c/T_c < 1/64$ ) will unequivocally determine what is the state of affairs. However, the width of a transition is not sufficient to assess the quality of a sample. Therefore, Table 1 could be a means of clarifying the influence of preparative conditions, microstructures, crystalline defects and stoichiometry on the superconducting transition temperature  $T_c$ . Even in the highest quality single crystal disordered domains are present that cannot be distinguished by X-ray spectroscopy or other measurement procedures.

**Table 1:** Universal set of critical temperatures  $T_c$ 

$\nu$	$\kappa$	$T_c$ [K]	$\Delta T_c / T_c$ min	$\nu$	$\kappa$	$T_c$ [K]	$\Delta T_c / T_c$ min	$\nu$	$\kappa$	$T_c$ [K]	$\Delta T_c / T_c$ min
1/9	$\alpha^2$	0.007	0.125	9	$\alpha^2$	0.541	0.047	1/6	$(2\pi)^{-1}$	30.0	0.039
1/8	$\alpha^2$	0.008	0.111	1/7	$\alpha/2$	0.589	0.082	8	$\alpha/2$	33.0	0.090
1/7	$\alpha^2$	0.009	0.125	1/6	$\alpha/2$	0.688	0.143	1/5	$(2\pi)^{-1}$	36.1	0.030
1/6	$\alpha^2$	0.010	0.143	1/5	$\alpha/2$	0.825	0.111	9	$\alpha/2$	37.1	0.029
1/5	$\alpha^2$	0.012	0.111	2/9	$\alpha/2$	0.917	0.100	2/9	$(2\pi)^{-1}$	40.1	0.073
2/9	$\alpha^2$	0.013	0.100	1/4	$\alpha/2$	1.031	0.111	1/4	$(2\pi)^{-1}$	45	0.111
1/4	$\alpha^2$	0.0150	0.111	2/7	$\alpha/2$	1.179	0.125	2/7	$(2\pi)^{-1}$	52	0.125
2/7	$\alpha^2$	0.017	0.125	1/3	$\alpha/2$	1.38	0.125	1/3	$(2\pi)^{-1}$	60.1	0.125
1/3	$\alpha^2$	0.0200	0.125	3/8	$\alpha/2$	1.55	0.067	3/8	$(2\pi)^{-1}$	68	0.067
3/8	$\alpha^2$	0.023	0.067	2/5	$\alpha/2$	1.65	0.063	2/5	$(2\pi)^{-1}$	72	0.063
2/5	$\alpha^2$	0.0240	0.063	3/7	$\alpha/2$	1.768	0.037	3/7	$(2\pi)^{-1}$	77.3	0.037
3/7	$\alpha^2$	0.0258	0.037	4/9	$\alpha/2$	1.83	0.036	4/9	$(2\pi)^{-1}$	80.1	0.036
4/9	$\alpha^2$	0.027	0.036	1/2	$\alpha/2$	2.06	0.111	1/2	$(2\pi)^{-1}$	90.1	0.111
1/2	$\alpha^2$	0.030	0.111	5/9	$\alpha/2$	2.29	0.029	5/9	$(2\pi)^{-1}$	100	0.029
5/9	$\alpha^2$	0.033	0.029	4/7	$\alpha/2$	2.357	0.028	4/7	$(2\pi)^{-1}$	103	0.028
4/7	$\alpha^2$	0.034	0.028	3/5	$\alpha/2$	2.48	0.042	3/5	$(2\pi)^{-1}$	108	0.042
3/5	$\alpha^2$	0.036	0.042	5/8	$\alpha/2$	2.58	0.040	5/8	$(2\pi)^{-1}$	113	0.040
5/8	$\alpha^2$	0.038	0.040	2/3	$\alpha/2$	2.75	0.063	2/3	$(2\pi)^{-1}$	120	0.062
2/3	$\alpha^2$	0.040	0.062	5/7	$\alpha/2$	2.95	0.050	5/7	$(2\pi)^{-1}$	129	0.050
5/7	$\alpha^2$	0.043	0.050	3/4	$\alpha/2$	3.09	0.037	3/4	$(2\pi)^{-1}$	135	0.037
3/4	$\alpha^2$	0.045	0.037	7/9	$\alpha/2$	3.21	0.029	7/9	$(2\pi)^{-1}$	140	0.029
7/9	$\alpha^2$	0.047	0.029	4/5	$\alpha/2$	3.300	0.028	4/5	$(2\pi)^{-1}$	144	0.028
4/5	$\alpha^2$	0.048	0.028	5/6	$\alpha/2$	3.438	0.029	5/6	$(2\pi)^{-1}$	150	0.029
5/6	$\alpha^2$	0.0501	0.029	6/7	$\alpha/2$	3.54	0.021	6/7	$(2\pi)^{-1}$	155	0.021
6/7	$\alpha^2$	0.0515	0.021	7/8	$\alpha/2$	3.61	0.0159	7/8	$(2\pi)^{-1}$	158	0.0159
7/8	$\alpha^2$	0.0526	0.0159	8/9	$\alpha/2$	3.667	<b>0.0156</b>	8/9	$(2\pi)^{-1}$	160	<b>0.0156</b>
8/9	$\alpha^2$	0.0534	<b>0.0156</b>	1	$\alpha/2$	4.125	0.111	1	$(2\pi)^{-1}$	180	0.111
1	$\alpha^2$	0.060	0.111	9/8	$\alpha/2$	4.64	0.0159	9/8	$(2\pi)^{-1}$	203	0.0159
9/8	$\alpha^2$	0.0676	0.0159	8/7	$\alpha/2$	4.71	<b>0.0156</b>	8/7	$(2\pi)^{-1}$	206	<b>0.0156</b>
8/7	$\alpha^2$	0.0687	<b>0.0156</b>	7/6	$\alpha/2$	4.81	0.020	7/6	$(2\pi)^{-1}$	210	0.020
7/6	$\alpha^2$	0.0701	0.020	6/5	$\alpha/2$	4.95	0.028	6/5	$(2\pi)^{-1}$	216	0.028
6/5	$\alpha^2$	0.072	0.028	5/4	$\alpha/2$	5.16	0.029	5/4	$(2\pi)^{-1}$	225	0.029
5/4	$\alpha^2$	0.075	0.029	9/7	$\alpha/2$	5.30	0.028	9/7	$(2\pi)^{-1}$	232	0.028
9/7	$\alpha^2$	0.077	0.028	4/3	$\alpha/2$	5.50	0.036	4/3	$(2\pi)^{-1}$	240	0.036
4/3	$\alpha^2$	0.080	0.036	7/5	$\alpha/2$	5.78	0.048	7/5	$(2\pi)^{-1}$	252	0.048
7/5	$\alpha^2$	0.084	0.048	3/2	$\alpha/2$	6.19	0.067	3/2	$(2\pi)^{-1}$	270	0.067
3/2	$\alpha^2$	0.090	0.067	8/5	$\alpha/2$	6.60	0.042	8/5	$(2\pi)^{-1}$	288	0.042
8/5	$\alpha^2$	0.096	0.042	5/3	$\alpha/2$	6.88	0.040	5/3	$(2\pi)^{-1}$	300	0.040
5/3	$\alpha^2$	0.1002	0.040	7/4	$\alpha/2$	7.22	0.029	7/4	$(2\pi)^{-1}$	315	0.029
7/4	$\alpha^2$	0.105	0.029	9/5	$\alpha/2$	7.43	0.028	9/5	$(2\pi)^{-1}$	324	0.028
9/5	$\alpha^2$	0.108	0.028	2	$\alpha/2$	8.25	0.100	2	$(2\pi)^{-1}$	361	0.100
2	$\alpha^2$	0.120	0.100	9/4	$\alpha/2$	9.282	0.037	9/4	$(2\pi)^{-1}$	406	0.037
9/4	$\alpha^2$	0.135	0.037	7/3	$\alpha/2$	9.63	0.036	7/3	$(2\pi)^{-1}$	421	0.036

$\nu$	$\kappa$	$T_c$ [K]	$\Delta T_c / T_c$ min	$\nu$	$\kappa$	$T_c$ [K]	$\Delta T_c / T_c$ min	$\nu$	$\kappa$	$T_c$ [K]	$\Delta T_c / T_c$ min
7/3	$\alpha^2$	0.1402	0.036	5/2	$\alpha/2$	10.31	0.067	5/2	$(2\pi)^{-1}$	451	0.067
5/2	$\alpha^2$	0.150	0.067	8/3	$\alpha/2$	11.0	0.063	8/3	$(2\pi)^{-1}$	481	0.062
8/3	$\alpha^2$	0.1603	0.062	3	$\alpha/2$	12.4	0.111	3	$(2\pi)^{-1}$	541	0.111
3	$\alpha^2$	0.180	0.111	7/2	$\alpha/2$	14.4	0.143	7/2	$(2\pi)^{-1}$	631	0.143
7/2	$\alpha^2$	0.210	0.143	4	$\alpha/2$	16.5	0.125	4	$(2\pi)^{-1}$	721	0.125
4	$\alpha^2$	0.240	0.125	9/2	$\alpha/2$	18.6	0.079	9/2	$(2\pi)^{-1}$	811	0.111
9/2	$\alpha^2$	0.270	0.111	1/9	$(2\pi)^{-1}$	20.0	0.030	5	$(2\pi)^{-1}$	901	0.100
5	$\alpha^2$	0.300	0.100	5	$\alpha/2$	20.6	0.029	6	$(2\pi)^{-1}$	1082	0.167
6	$\alpha^2$	0.361	0.167	1/8	$(2\pi)^{-1}$	22.5	0.085	7	$(2\pi)^{-1}$	1262	0.143
7	$\alpha^2$	0.421	0.090	6	$\alpha/2$	24.8	0.040	8	$(2\pi)^{-1}$	1442	0.125
1/9	$\alpha/2$	0.458	0.049	1/7	$(2\pi)^{-1}$	25.8	0.039	9	$(2\pi)^{-1}$	1622	0.111
8	$\alpha^2$	0.481	0.047	7	$\alpha/2$	28.9	0.040				
1/8	$\alpha/2$	0.516	0,049								

**Note:** The dimensionless transition width  $\Delta T_c/T_c$  indicates the smallest temperature difference to the neighboring critical temperatures. The inequality  $1/64 \leq \Delta T_c/T_c \leq 1/6$  applies.

**Table 2:** Elements

<b>Element</b> <sup>a)</sup>	<b>experiment</b> [K]	<b>Conj</b> <sup>b)</sup> [K]	<b>T<sub>c</sub><sup>χ</sup></b> <sup>c)</sup> [K]	<b>T<sub>c</sub><sup>C</sup></b> <sup>d)</sup> [K]	<b>T<sub>c</sub><sup>P</sup></b> <sup>e)</sup> [K]	<b>ΔT<sub>c</sub><sup>P</sup></b> <sup>f)</sup> [K]	<b>varia</b>	<b>ref</b>
W morphol. unknown	0.0152(1) <sup>g)</sup>	0.0150						[6]
Be morphol. unknown	0.0201(1) <sup>g)</sup>	0.0200						[6]
Be	0.0244	0.0240	x					[7]
Be morphol. unknown	0.026	0.0258	x					[8]
Ir morphol. unknown	0.0990(5) <sup>g)</sup>	0.1002						[6]
Ir polycr. melt	0.140	0.1402	x					[9]
Pa <sup>231</sup>	0.430	0.421	x					[10]
Ti	0.480	0.481	x					[11]
Cd	0.518	0.516		x				[12]
Ti morphol. unknown	0.511(5)	0.516	x					[13]
Ru polycr. melt	0.509(2)	0.516		x				[14]
Os polycr. melt	0.71(2)	0.69	x					[15]
Zn	0.825	0.825		x				[16]
Mo	0.917	0.917		x				[17]
	0.830	0.825						
α-Ga polycr.	1.078	1.031		x				[12]
Al morphol. unknown	1.178(2) <sup>g)</sup>	1.179						[6]
Th polycr. melt	1.374(1)	1.375		x				[18]
SIC-Pb / Si(111) epitaxial	1.83	1.83					STM	[19]
Tl <sup>205</sup>	2.386(1)	2.357	x					[20]
Cr / MgO(001) epitaxial	3.20	3.21			x		50nm	[21]
Pd irradiated	3.20	3.21			x		T <sub>c</sub> <sup>max</sup>	[22]
In <sup>113</sup> polycr. melt	3.433	3.438	x					[23]
In	3.431(1)	3.438	x				Tl	
Sn <sup>123</sup>	3.6669	3.6669	x					[24]
Sn <sup>124</sup>	3.666(3)	3.667	x				Tl	
α-Hg <sup>203.4</sup> morphol. unknown	4.126	4.125	x					[25]
dhcp-La polycr.	5.04	4.95		x				[26]
V	5.13	5.16	x					[27]
V zone melting	5.31(4)	5.30	x					[28]
Pb / Si(111) epitaxial	5L	5.75	5.78		5.9	0.3	double coil	[29]
β-Ga nanowire array	6.2	6.19	x					[30]
Pb	7.191(2)	7.219					μSR	[31]
Pb porous glass nanocomposite	7.22	7.219	x					[32]
bcc-Te phase boundary	7.4	7.43			x		@ 35 GPa	[33]
Nb	9.288(2)	9.282	x					[34]
β-Zr polycr. foil @ 30 GPa	11.0	11.0			x		T <sub>c</sub> <sup>max</sup>	[35]

**Table 3:** Binaries

<b>Compound</b> <sup>a)</sup>	<b>experiment</b>	<b>Conj</b> <sup>b)</sup>	<b>T<sub>c</sub><sup>X</sup></b> <sup>c)</sup>	<b>T<sub>c</sub><sup>C</sup></b> <sup>d)</sup>	<b>T<sub>c</sub><sup>P</sup></b> <sup>e)</sup>	<b>ΔT<sub>c</sub><sup>P</sup></b> <sup>f)</sup>	<b>varia</b>	<b>ref</b>
	[K]	[K]	[K]	[K]	[K]	[K]		
AuAl <sub>2</sub> morphol. unknown	0.1607(5) <sup>g)</sup>	0.1603						[6]
AuIn <sub>2</sub> morphol. unknown	0.208(1) <sup>g)</sup>	0.210						[6]
TaSi <sub>2</sub>	0.345(10)	0.361	0.345(10)	0.353(3)				[36]
TiO/MgO(001) epitaxial	0.45	0.458			x			[37]
UPt <sub>3</sub>	0.514	0.516			x		doublet	[38]
	0.460	0.458					figure 4.2	
UPt <sub>3</sub>	0.543(2)	0.541		x			doublet	[39]
	0.489(2)	0.481						
WP	0.84	0.825	0.84		0.85	0.02		[40]
NbSe <sub>2</sub> /6H-SiC epitaxial	1L	0.920	0.917				STS	[41]
CoSi <sub>2</sub>	1.03	1.031			x			[42]
CaSn <sub>3</sub>	1.178	1.179			x			[43]
NbO polycr. melt	1.38	1.38	x					[44]
Ba <sub>24</sub> Si <sub>100</sub> polycr.	1.55	1.55	x					[45]
PdTe <sub>2</sub>	1.64	1.65	x					[46]
SrC <sub>6</sub>	1.65	1.65	1.65	1.65				[47]
UTe <sub>2</sub>	S2	1.68(3)	1.65		x			[48]
	S3	1.77(3)	1.77					
	S4	1.85(3)	1.83					
	S6	2.00(4)	2.06					
UTe <sub>2</sub>		2.05	2.06		2.05	x		[49]
OsB <sub>2</sub>		2.06	2.06	2.06	2.1			[50]
NbGe <sub>2</sub>		2.06	2.06	x	2.05	0.07		[51]
WB <sub>4.2</sub> polycr.		2.05	2.06	2.0	2.05	2.25	0.2	[52]
AuSn <sub>4</sub>		2.35	2.36		x			[53]
LiBi		2.48	2.48	2.38	2.48			[54]
Bi-metal graphite		2.48(2)	2.48	x				[55]
Fe <sup>10</sup> B <sub>4</sub> polycr.		2.95	2.95	x				[56]
2H-TaS <sub>2</sub> exfoliation	2L	3.0	2.95	x				[57]
	3L	2.5	2.48					
	5L	2.05	2.06					
Pb <sub>2</sub> Pd		2.95	2.95	2.85(10)	2.95	3.0	0.1	[58]
LaAl <sub>2</sub>		3.301(2)	3.300	x				[59]
BeAu polycr.		3.3	3.30	3.3	3.3	3.3	0.05	[60]
KBi <sub>2</sub>		3.57	3.54	3.52	3.57	3.55	0.1	[61]
PdBi <sub>1-x</sub>		3.5	3.54	3.5		3.5	0.15	x=0.07 [62]
CaSn <sub>3</sub>		4.15	4.13	4.15		4.2	0.2	[63]
IrGe polycr.		4.74(3)	4.71	4.74	4.6	4.7(1)	0.1	[64]
β-Bi <sub>2</sub> Pd		4.95(5)	4.95		4.95(5)	5.0	0.1	[65]
β-Bi <sub>2</sub> Pd		4.9	4.95		4.8	5.0	0.02	[66]
IrGe polycr.		5.17	5.16		5.17	5.5	0.5	[67]
2H-NbSe <sub>2</sub>		6.90	6.88			x		[68]
2H-NbSe <sub>2</sub>		7.20	7.22			7.20		[69]
TaC		10.3	10.31	10.3	9.7	10.7	0.4	[70]
NbN/SiC epitaxial		16.5	16.50			x	50nm	[71]

**Table 4:** Complex compounds

Compound <sup>a)</sup>	experiment [K]	Conj <sup>b)</sup> [K]	$T_c^{\chi}$ <sup>c)</sup> [K]	$T_c^C$ <sup>d)</sup> [K]	$T_c^P$ <sup>e)</sup> [K]	$\Delta T_c^P$ <sup>f)</sup> [K]	varia	ref
Ag <sub>5</sub> Pb <sub>2</sub> O <sub>6</sub>	0.0524	0.0526	x					[72]
CePt <sub>3</sub> Si	0.46	0.458		0.46	0.8	0.05		[73]
Sr <sub>2</sub> RuO <sub>4</sub>	1.03 0.050	1.031 0.0501					mag. rel.	[74]
LaTr <sub>2</sub> Al <sub>20</sub>	Tr = Ta Tr = Nb Tr = Ti Tr = V	1.03(2) 1.05(2) 0.46(1) 0.15(2)	1.031 1.031 0.458 0.150	x x x x				[75]
YNiSi <sub>3</sub>		1.36(3)	1.38	1.36(3)	1.35(5)	1.42(2)		[76]
A <sub>x</sub> V <sub>2</sub> Al <sub>20</sub>	polycr. A <sub>x</sub> =Ge <sub>0.2</sub> A <sub>x</sub> =Y	1.66 0.69	1.65 0.688	x x				[77]
(Au <sub>1-x</sub> Pd <sub>x</sub> )Ga <sub>2</sub>	polycr. melt	1.773(20)	1.768	x			x=0.07	[78]
CeCoIn <sub>5</sub>		2.3	2.29		x			[79]
PuCoIn <sub>5</sub>		2.5	2.48		x			[80]
SnTaS <sub>2</sub>		2.97	2.95	2.97	2.88	3.00	0.05	[81]
LiGa <sub>2</sub> Ir	polycr.	2.95	2.95	2.95	2.94	3.0	0.05	[82]
La <sub>3</sub> T <sub>4</sub> Sn <sub>13</sub>	T = Rh T = Ir	3.1 2.5	3.09 2.48	3.1 2.5	3.1 2.5			[89]
Ba(Ni <sub>1-x</sub> Cu <sub>x</sub> ) <sub>2</sub> As <sub>2</sub>		3.20	3.21	3.2	3.2		x <sub>opt</sub> ≈0.292	[83]
LaTr <sub>2</sub> Al <sub>20</sub>	Tr = Mo Tr = W	3.22 1.81	3.21 1.83	x x				[84]
CaIrSi <sub>3</sub>		3.3	3.30		x			[85]
BaNi <sub>2</sub> (As <sub>1-x</sub> P <sub>x</sub> ) <sub>2</sub>	phase boundary	3.3 0.7	3.30 0.69		3.3 x	3.35	0.05	x <sub>opt</sub> ≈0.077 x≈0.07
TlNi <sub>2</sub> Se <sub>2</sub>		3.7	3.67	3.7	3.7	3.7	0.05	[87]
Bi <sub>2</sub> PdPt		4.0(1)	4.13	3.99(2)		4.0(1)	0.1	[88]
Sr <sub>3</sub> T <sub>4</sub> Sn <sub>13</sub>	T = Rh T = Ir	4.15 4.95	4.13 4.95	4.2 5.0	4.15 4.95			[89]
Sc <sub>5</sub> Rh <sub>6</sub> Sn <sub>18</sub>		4.8 5.15	4.81 5.16	5.0 4.97	4.8 4.97			[90] [91]
Y <sub>7</sub> Ru <sub>4</sub> InGe <sub>12</sub>		5.8	5.78	5.8		5.8	0.2	[92]
DyNi <sub>2</sub> B <sub>2</sub> C		6.2(1)	6.19	6.1		6.4	0.4	[93]
MgCNi <sub>3</sub>		6.85	6.88		x			[94]
β-ThRh <sub>1-x</sub> Ir <sub>x</sub> Ge	polycr.	6.88	6.88	x			x <sub>opt</sub> =0.5	[95]
W <sub>7</sub> Re <sub>13</sub> B	polycr.	7.22(2)	7.22	7.22(2)		7.3	0.1	[96]
LaRu <sub>4</sub> P <sub>12</sub>	powdered	7.20	7.22	x				[97]
Mo <sub>5</sub> Ge <sub>2</sub> P	polycr.	8.24	8.25		x			[98]
LaRu <sub>4</sub> As <sub>12</sub>	polycr.	10.3	10.31		10.3	10.6	0.6	[99]
TmNi <sub>2</sub> B <sub>2</sub> C	isotopically enriched B	11 g)	11.0					[100]
LiTi <sub>2</sub> O <sub>4</sub> / MgAl <sub>2</sub> O <sub>4</sub>	epitaxial	11.00(25)	11.00				tunnelling	[101]
Fe <sub>1+y</sub> Te <sub>0.6</sub> Se <sub>0.4</sub>		14.5	14.4		x		T <sub>c</sub> <sup>opt</sup>	[102]
PuCoGa <sub>5</sub>		18.5	18.6	18.5	18.5	18.6	0.4	[103]

**Table 5:** Superconductors with  $T_c \geq 20$  K

Compound <sup>a)</sup>	experiment [K]	Conj [K]	$T_c^\gamma$ <sup>c)</sup> [K]	$T_c^C$ <sup>d)</sup> [K]	$T_c^P$ <sup>e)</sup> [K]	$\Delta T_c^P$ <sup>f)</sup> [K]	varia	ref
K <sub>3</sub> C <sub>60</sub>	19.8	20.0			19.8	0.2		[104]
BaFe <sub>2-x</sub> Ni <sub>x</sub> As <sub>2</sub>	20.5	20.0			20.5	1	$x_{opt} \approx 0.096$	[105]
		20.6						
YPd <sub>5</sub> B <sub>3</sub> C <sub>0.3</sub> polycr.	22.6	22.5	22.6		23	1		[106]
Ba <sub>1-x</sub> Rb <sub>x</sub> Fe <sub>2</sub> As <sub>2</sub>	22.6(2)	22.5	x				$x_{opt} \approx 0.1$	[107]
Ba(Fe <sub>1-x</sub> Co <sub>x</sub> ) <sub>2</sub> As <sub>2</sub>	25.5	25.8			25.5	0.7	$x_{opt} \approx 0.063$	[108]
BaFe <sub>2</sub> (As <sub>1-x</sub> P <sub>x</sub> ) <sub>2</sub>	29.2	28.9		x			$x_{opt} \approx 0.3$	[109]
	30.0	30.0	30.0		30.0	0.4	$x_{opt} \approx 0.33$	[110]
CsCa <sub>2</sub> Fe <sub>4</sub> As <sub>4</sub> F <sub>2</sub>	30.0	30.0	29.4		30.0	0.6		[111]
CaKFe <sub>4</sub> As <sub>4</sub> polycr.	33.2	33.0	x					[112]
EuRbFe <sub>4</sub> As <sub>4</sub>	36.5	36.1	36.5		36.7	0.4		[113]
MgB <sub>2</sub> polycr. grains	37.25 mean	37.13	x					[114]
		37.1	x					[115]
MgB <sub>2</sub> /Al <sub>2</sub> O <sub>3</sub> epitaxial	40.2	40.1			40.2	0.15	1.3 μm	[116]
SmFeAsO <sub>0.8</sub> F <sub>0.2</sub>	45(1)	45.1	x					[117]
PrFeAsO <sub>1-y</sub>	45	45.1	x					[118]
Li <sup>+</sup> - FeSe nanoflake	45.0	45.1			x		$T_c^{opt}$	[119]
PdH <sub>x</sub> morphol. unknown	52	52			x		figure 4-7	[120]
NdFeAsO <sub>1-y</sub> polycr.	53	52	x				$T_c$ saturation	[121]
	33	33.0	x				$T_c$ boundary	
YBa <sub>2</sub> Cu <sub>3</sub> O <sub>6+x</sub>	61(2) <sup>g)</sup>	60.1					$x \approx 0.60$	[122]
	90(1) <sup>g)</sup>	90.1					$x=1$	
PdD <sub>x</sub> morphol. unknown	60	60.1			x		figure 4-8	[120]
PrBa <sub>2</sub> Cu <sub>3</sub> O <sub>x</sub>	80	80.1	x					[123]
YBa <sub>2</sub> Cu <sub>3</sub> O <sub>7-δ</sub> single phase	90 <sup>g)</sup>	90.1					$T_c^{max}$	[124]
Bi <sub>2</sub> Sr <sub>2</sub> CaCu <sub>2</sub> O <sub>8+δ</sub>	90	90.1	x				$T_c^{max}$	[125]
	77	77.3					$T_c^{min}$	
FeSe/SrTiO <sub>3</sub> (001) epitaxial	1L	100	100		100	1	figure S4	[126]
		109	108	x			figure 3b	
(Cu,C)Ba <sub>2</sub> Ca <sub>3</sub> Cu <sub>4</sub> O <sub>11+δ</sub>	112	113	112		111	1.4		[127]
Hg-1223 polycr.	135 <sup>g)</sup>	135						[128]
Hg-1234 polycr.	120 <sup>g)</sup>	120					annealed	
Hg-1212 polycr.	104 <sup>g)</sup>	103					as prepared	
Hg-1256 polycr.	100 <sup>g)</sup>	100					annealed	
Hg-1245 polycr.	101 <sup>g)</sup>						as prepared	
SH <sub>x</sub> morphol. unknown	203(1)	203	203(1)				figure 4a	[129]
@ 155 GPa								
LuH <sub>3-δ</sub> N <sub>ε</sub> morphol. unknown	269	270	269		269	2	figure 12	[130]
@ 16kbar								
thin film resistor amorphous	408 <sup>h)</sup>	406			408	<5	figure 16.23	[1c]

## Notes to the tables 2 – 5

- a) Samples are single crystals unless otherwise noted.
- b) The corresponding  $v$ ,  $\kappa$  and  $\Delta T_c / T_c$  can be found in Table 1.
- c) Defined as the onset of the diamagnetism.
- d) Determined by specific heat using an equal-area entropy construction.
- e) Defined as the temperature  $T_c^P$  (onset) at which  $\rho$  begins to deviate from the normal-state behavior.
- f) The width  $\Delta T_c$  of the superconducting transition given by the difference  $T_c^P$  (onset) and  $T_c^P$  ( $\rho=0$ ).
- g) Measurement method unknown.
- h) Transient, exceptionally low resistivity in a four-point measurement.

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